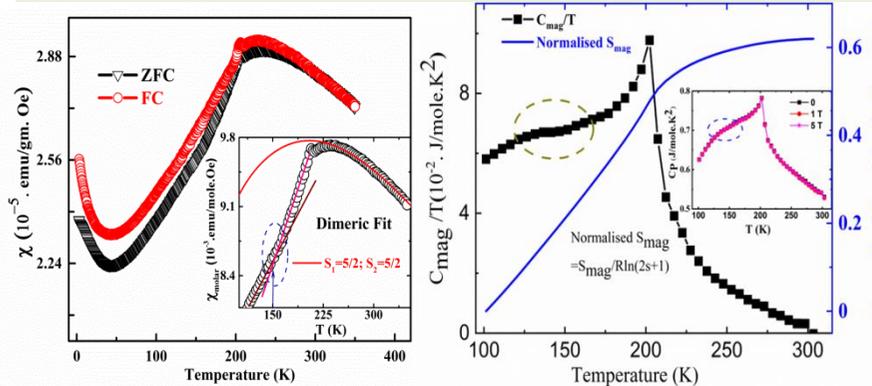


Abstract

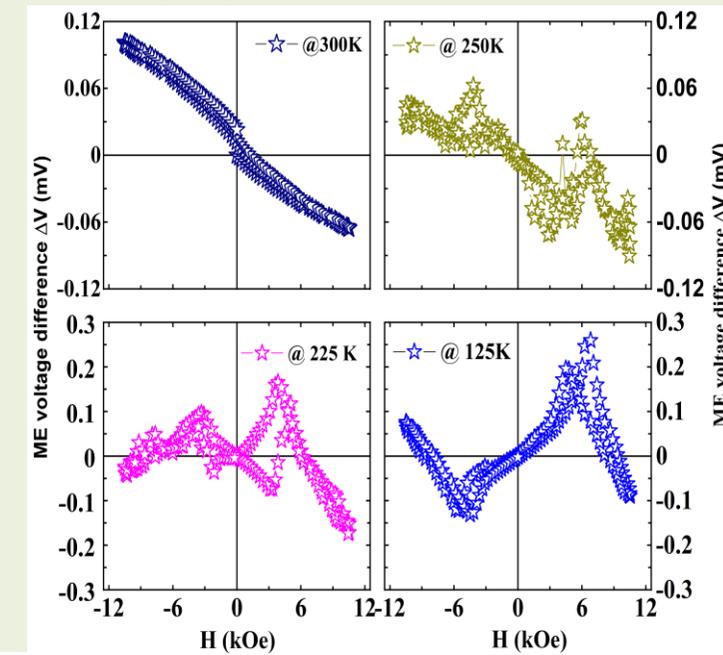
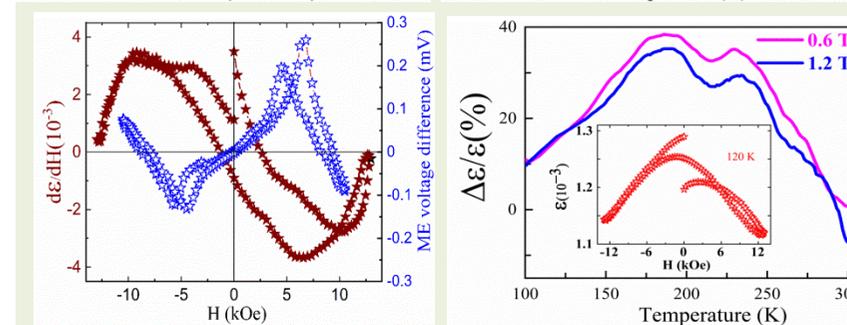
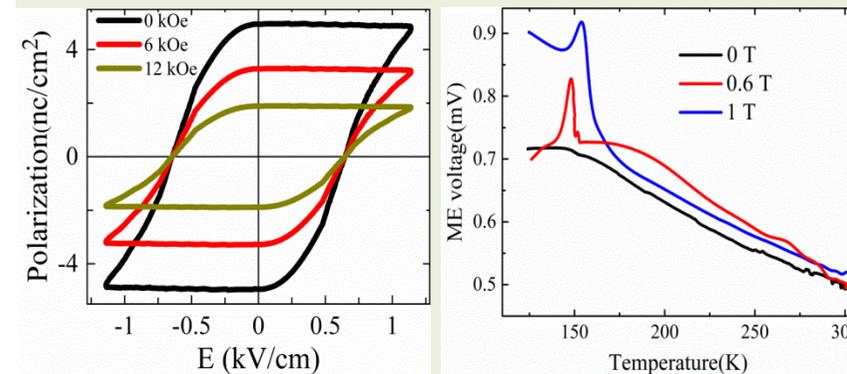
We present the rare existence of d^5 off-centering induced weak ferroelectric polarization and magnetoelectric (ME) properties in the G type ($T_N \sim 210$ K) antiferromagnet Fe_2TeO_6 (FTO) compound. The origin of ferroelectricity (FE) is associated with both lattice and asymmetric electron density distribution around the ion cores. ME coupling is observed in magnetic field-dependent polarization, ME voltage, and magnetostrain measurements. Short-range magnetic ordering due to intrabilayer dimeric exchange coupling via the double oxygen bridged Fe-O1-Fe pathway is proposed to play a dominating role to exhibit the negative nonlinear magnetic field dependent ME behavior at 300 K. Interbilayer exchange via Fe-O2-Fe pathways dominantly determines the hysteretic nonlinear magnetic field dependent ME response below T_N . The observed nonlinear ME coupling signifies magnetoelasticity as manifested in the temperature and magnetic field-dependent strain measurement. Hence the rare existence of FE and magnetoelectric coupling by d^5 ion is presented in FTO.

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Macroscopic magnetic and magnetoelectric properties

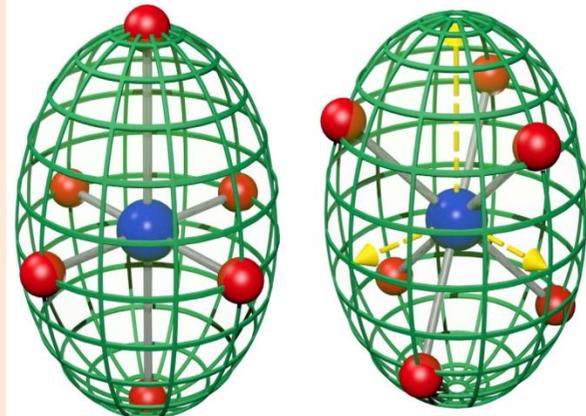
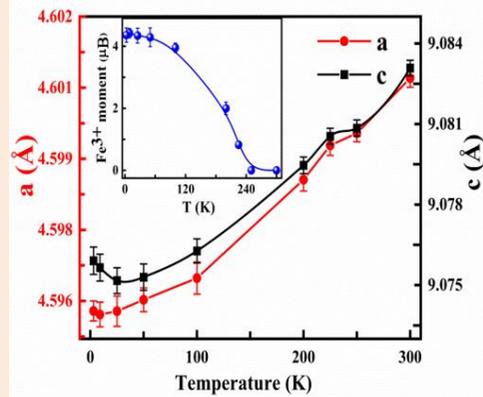
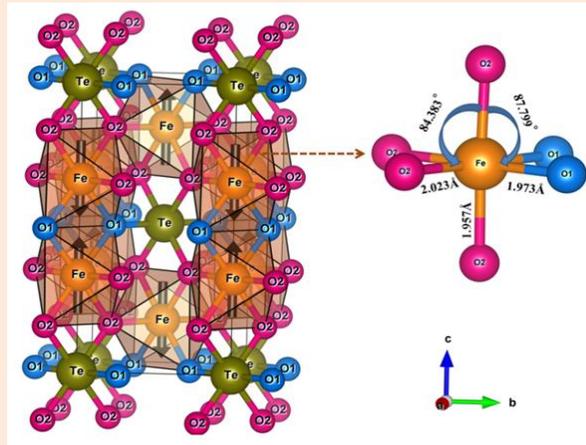
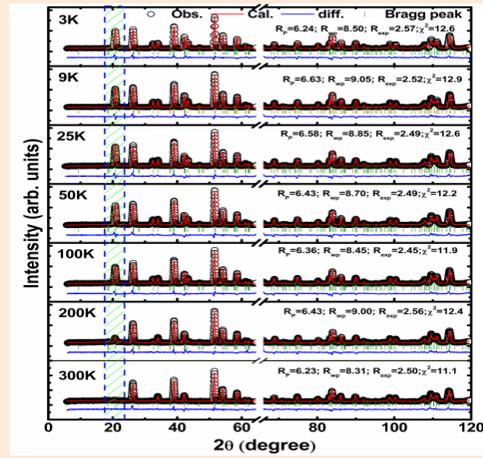


- $T_N \sim 210$ K,
- $5/2-5/2$ dimeric interaction: Magnetic correlation above T_N
- Direct evidences of magnetoelectric (ME) Coupling; Direct Measurements:
 - Polarization vs Electric field under magnetic field at 300 K (RT)
 - ME voltage : Different kind of ME coupling at $T_N < T < T_N$
 - Magnetostrain: Magnetoelastic coupling



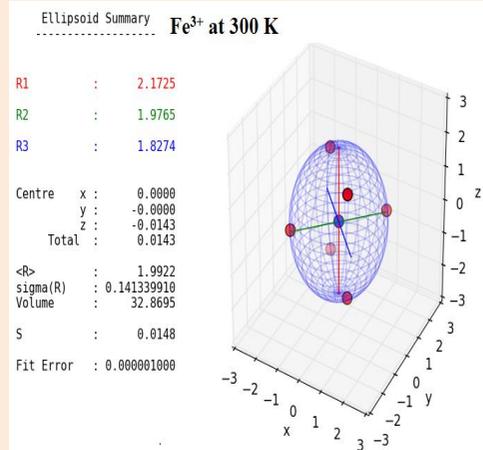
Microscopic understanding of ferroelectricity and ME coupling

Neutron diffraction study and Ellipsoid analysis



Fitting the coordination polyhedra with : Minimum Bounding Ellipsoid (MBE) : Bond distortion & Angular distortion [1]

- i) Principal radii; R_1, R_2, R_3 ($R_1 \geq R_2 \geq R_3$)
- ii) d : Central cationic displacement
- iii) $\langle R \rangle$: Mean radius; Size
- iv) $\sqrt{\sigma^2(R)}$; Variance: distortion
- v) $S = (R_3/R_2 - R_2/R_1)$; $S < 0$: Oblate, $S > 0$: Prolate, $S = 0$: Sphere

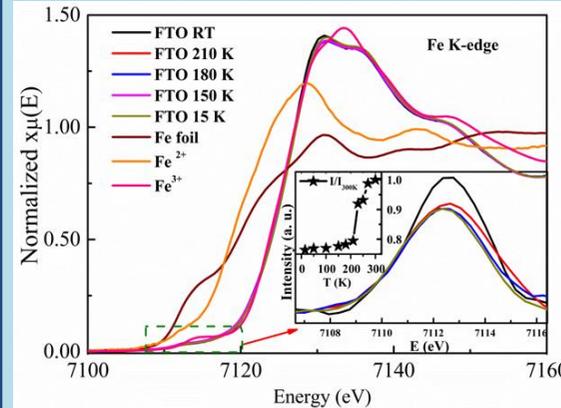


- Reduced (13%) moment of Fe^{3+} ; $4.35 \mu_B$
- d^5 off centering of Fe^{3+}
- Deviation from D_{4h} symmetry
- Change in polyhedra shape and size at T_N : Magnetostructural correlation
- Nonmonotonic temperature variation of distortion
- Non-octahedral hybridization

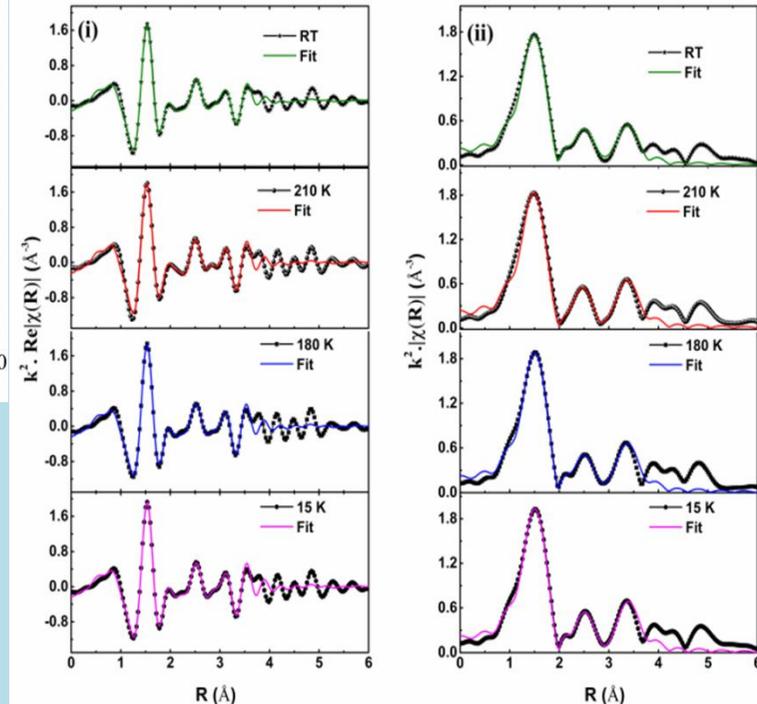
X-ray Absorption Spectroscopy

XANE: X-ray near edge spectra

EXAFS: Extended X-ray absorption fine structure



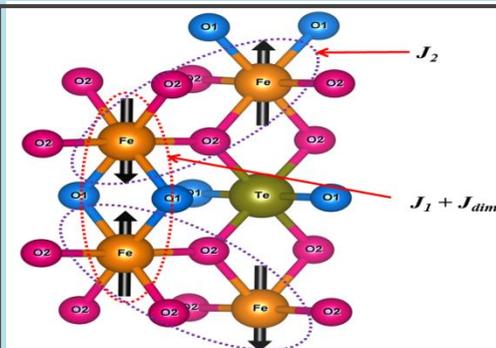
- ❑ Fe K edge confirms existence of Fe^{3+}
- ❑ Low intense Gaussian peak; Small deviation from D_{4h} symmetry
- ❑ Sudden fall in pre-edge intensity at 210 K ; Fe-O hybridization depends upon magnetic correlation [2, 3]



$$\chi(E) = \sum_j \frac{N_j}{kR_j} |f_j(k)| e^{-\frac{R_j}{\lambda_e}} e^{-2k^2\sigma_j^2} \sin(2kR_j + 2\delta_1 + \arg(f_j(k)))$$

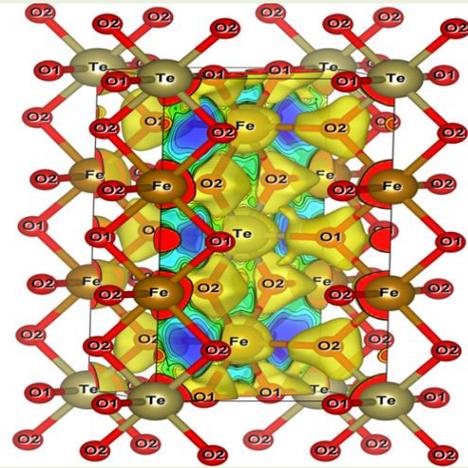
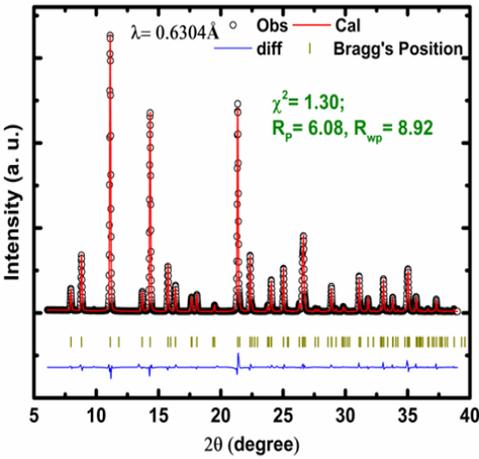
N_j : Coordination Number, R_j : Distance between scatterer & absorber, σ_j : Debye Waller factor; Represent disorder or deviation

- ❑ First coordination cell : D_{4h} symmetry and $P4_2/mnm$
- ❑ Fe-O (Apical) : 1.995(3) and σ^2 (Å^2): 0.0042
- ❑ Fe-O2 (Equatorial): 2.0032 ; σ^2 (Å^2): 0.006(3)
- ❑ Second coordination Fe-O: 3.795 (3) ; large σ^2 (Å^2): 0.0073 implies larger flexibility [4, 5]



Microscopic understanding of ferroelectricity and ME coupling

Charge density distribution calculation : MEM analysis



$$F(h) = \sum_r \rho(r) e^{-2i\pi h \cdot r}$$

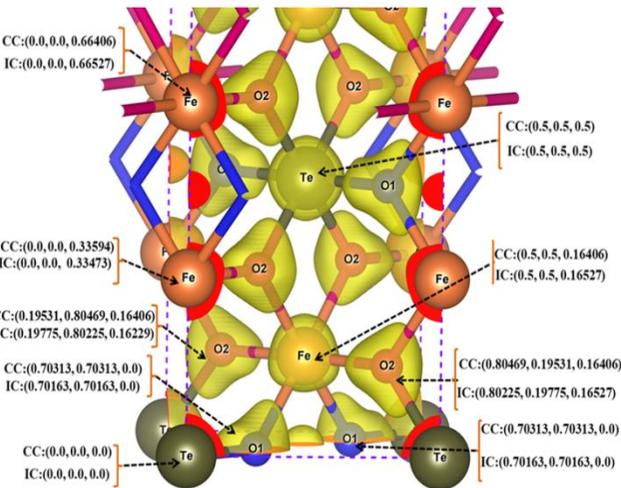
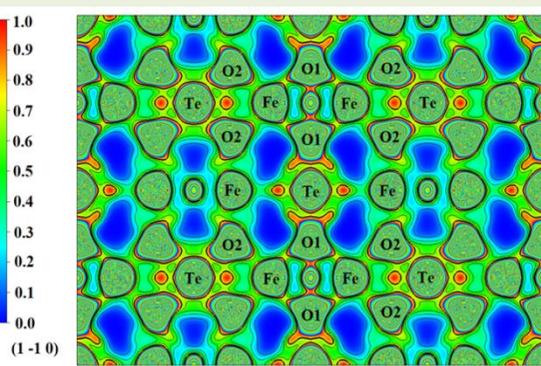
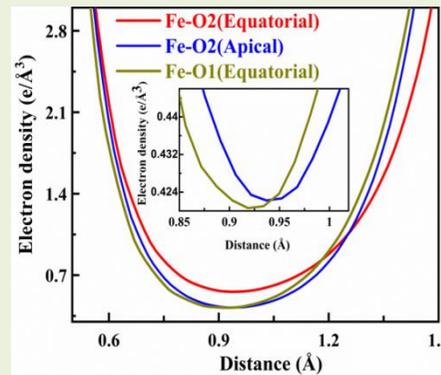
$$\rho(r) = \sum_h F_h e^{2i\pi h \cdot r}$$

Conventional Fourier method : Limited number of structure factor

- Unphysical negative density due to erroneous termination of calculation [6, 7]

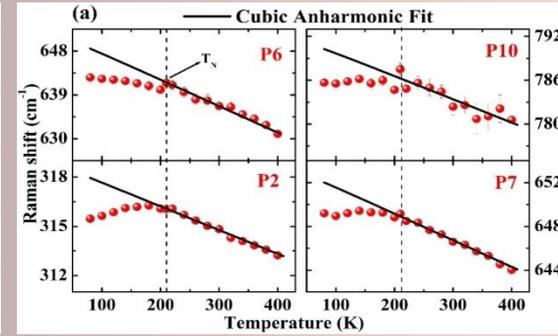
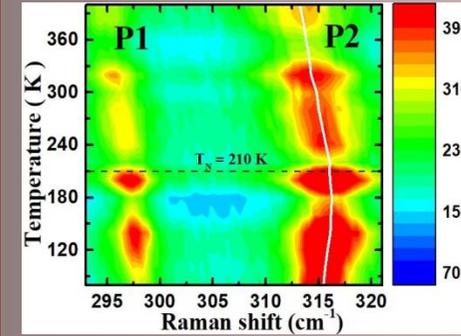
Maximum Entropy Method : MEM analysis

- ❖ 1d electron density : different Fe-O bonds
- ❖ Saddle point and density at saddle point are different
- ❖ Distortion in Fe^{3+} polyhedra



- Positions of the ionic centers (IC) and the charge cloud centers (CC) : Non-overlapping : ferroelectric polarization
- Microscopic polarization is estimated from the shift (Δc) of each IC and CC center
- $P = \frac{ne\Delta c}{V}$; n is the number of electrons, e is the electronic charge, and V is the volume of the unit cell.
- Electronic polarization $P \sim 0.05 \mu C/cm^2$

Raman spectroscopy: Spin phonon coupling



$$\omega_{anh}(T) = \omega(0) + C \left[1 + \frac{2}{\frac{\hbar\omega}{e^{2kT}} - 1} \right] \quad \Delta\omega_{sp-ph} = \omega_{exp}(T) - \omega_{anh}(T) = \lambda_{sp} \langle S_i \cdot S_j \rangle [8]$$

- ✓ Unequal Fe-O2 bond lengths of the Fe-O2-Fe unit give rise to asymmetric vibrations of phonon modes.

Conclusion

- AFM ordering below 210 K : Sublattice moment $\sim 4.35 \mu_B$ (reduced by $\sim 13\%$)
- Finite d^5 off-centering with the non D_{4h} symmetry in Fe^{3+} polyhedra.
- The shape, size, and off-centering displacement of Fe^{3+} varies non-monotonically and follow the magnetic transition
- Non-octahedral type of hybridization in Fe^{3+} polyhedra.
- Dependence of p-d hybridization on magnetic ordering
- Asymmetric electron density around the Fe^{3+} is realized. Hence the local distortion in Fe^{3+} is verified
- So both d^5 off-centering of Fe^{3+} along with electronic contribution result in finite polarization $\sim 0.05 \mu C/cm^2$

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